NANOSCALE STRUCTURES GENERATION WITHIN THE SURFACE LAYER OF METALS WITH SHORT UV LASER PULSES

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Abstract  
We have completed modeling of a laser pulse influence on a gold target. We have applied a hybrid atomistic-continuum model to analyze the physical mechanisms responsible for the process of nanostructuring. The model combines the advantages of Molecular Dynamics and Two Temperature Model. We have carried out a direct comparison of the modeling results and experimental data on nano-modification due to a single ps laser pulse at the energy densities significantly exceeding the melting threshold. The experimental data is obtained due to a laser pulse irradiation at the wavelength of 248 nm and duration of 1.6 ps. The mask projection (diffraction grating) creates the sinusoidal intensity distribution on a gold surface with periods of 270 nm, 350 nm, and 500 nm. The experimental data and modeling results have demonstrated a good match subject to complex interrelations between a fast material response to the laser excitation, generation of crystal defects, phase transitions and hydrodynamic motion of matter under condition of strong laser-induced non-equilibrium. The performed work confirms the proposed approach as a powerful tool for revealing the physical mechanisms underlying the process of nanostructuring of metal surfaces. Detailed understanding of the dynamics of these processes gives the possibility for designing the topology of functional surfaces on nano- and micro-scales.

Keywords  
UV laser pulses, nanostructuring, simulations, molecular dynamics

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ОБРАЗОВАНИЕ НАНОРАЗМЕРНЫХ СТРУКТУР В ПРИПОВЕРХНОСТНОМ СЛОЕ МЕТАЛЛОВ ПРИ ВОЗДЕЙСТВИИ УЛЬТРАКОРОТКОГО ЛАЗЕРНОГО ИМПУЛЬСА УЛЬТРАФИОЛЕТОВОГО ДИАПАЗОНА

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Язык статьи — английский
Образование наноразмерных структур в приповерхностном слое металлов...
induced melting, spallation, and ablation of thin and thick metal targets [16–18].

This work is a continuation of our previously published results [19, 20], where the mechanism of periodic nanostructuring process was studied both experimentally and theoretically with UV (248 nm) single laser pulses (with a periodic sinusoidally-shaped spatial intensity profile on the surface of a thick Au target) with the fluences just above the damage threshold. In the present work, however, the applied incident fluences ranged from near to significantly above the damage threshold valuing of 125, 160, and 250 mJ/cm². Experiments and simulations are executed with the same spatial laser intensity period of 270 nm. The spatial irradiation geometry of the experiment combined with a properly selected model volume gives the opportunity to obtain a direct comparison of experiment and simulation of nanostructuring of a thick Au target.

To obtain smaller structures, the use of ultrashort UV pulses instead of visible light pulses is advantageous since the shorter wavelength allows a far better spatial resolution. Meanwhile, the use of pulse length shorter than the characteristic electron-phonon relaxation time of the material (about 25 ps for gold at the fluencies near the melting threshold [21, 22]) allows for a complete deposition of the pulse energy in the electronic system before transferring it to the lattice, thus ensuring highly localized material modifications [23]. For the generation of deterministic sub-micron periodic patterns various irradiation concepts have been demonstrated. The required interference of two or more beams has been accomplished by mask projection [24], a two grating interferometer [25], or a proximity phase mask set-up [26]. A wide range of applications through optimized light coupling in photovoltaic [27], Light Emitting Diodes (LED) [28], and detector devices [29] can benefit from the gained knowledge and allow a detailed control of the fabrication process of next generation devices.

The Experiment

Similarly to [19, 20], all surface structuring experiments are accomplished by single pulse exposure using a combination of interference and mask projection to form a large area sinusoidal intensity distribution on the sample surface (Fig. 1). The applied two-beam interference (resulting in the sinusoidal pattern) allows reaching the highest possible spatial intensity gradient, crucial for keeping the simulated volume as small as possible. The smallest available modulation period \( d \) is then only limited by the numerical aperture (NA) of the applied optics and the laser wavelength \( \lambda \) and can be expressed as \( d = \lambda/2NA \). Thus, using a short wavelength of 248 nm and a large numerical aperture of NA = 0.4, a lower limit for the periodicity of approximately 270 nm is obtained (as compared to our previous experiments 310 nm). The used laser system consists of a frequency tripled Ti:Sa system seeding a KrF excimer amplifier producing pulse energies of up to 30 mJ at a wavelength of 248 nm and a repetition rate of 10 Hz. The pulse length of 1.6 ps is measured by using a frequency resolved optical gating (FROG) trace obtained with a device described in [30]. Single pulses are selected by a shutter. The used pulse energy is set by a variable attenuator. The flat top central part of the laser beam illuminates the mask comprising a circular aperture of 1 mm diameter in contact with a Cr-on-quartz grating of lines and spaces with 25 \( \mu m \) period and duty cycle 0.5. The demagnification of the Schwarzschild-objective used for mask imaging is adjusted to be 25°. Consequently, the diameter of the overall projected area on the workpiece is 40 \( \mu m \). Since all diffracted orders emerging from the transmission grating except the two first orders are blocked, the image is formed via two-beam interference, resulting in a sinusoidal intensity distribution across the 40 \( \mu m \) area. A schematic view of this distribution is shown in Fig. 1, b. The resulting structures, obtained with the experimental set-up described above, can be seen as Scanning Electron Microscopy (SEM) images in Fig. 2 for different incident fluences. The presented images reveal the periodic frozen structures with difference aspect ratio depending on the applied fluence. One can also see that with the increase of fluence, the number of opened voids to the surface grows as well. The mechanism of these periodic structures formation, generation and evolution of voids, and the microstructures character will be investigated with an advance theoretical approach described below in the next section.

![Fig. 1. Mask projection setup for a periodic surface structuring of an area of 40 \( \mu m \). A sinusoidal interference pattern with a period of 270 nm is obtained by using only the ± 1st diffraction orders (a). Schematic shape of the incident fluence distribution (\( F_{inc} \)) created by an ideal single laser pulse self-interference of two beams under a certain angle (b)](image)
Fig. 2. SEM pictures showing surface modifications from a single laser pulse with periodic line spacing of 270 nm corresponding to the periodic irradiation pattern created by two beam interference for the incident fluences of 125 mJ/cm² (a), 150 mJ/cm² (b), 175 mJ/cm² (c), and 200 mJ/cm² (d). Small segments of a larger structured area on a thick gold sample at four different average incident fluences are shown. The sample was tilted in the SEM by 45° relative to the electron beam axis to increase the contrast. The figure is taken from Ref. [19].

The Model

The essential concepts and applications of the combined atomistic-continuum model MD-TTM to study the evolution of metallic solids excited by an ultrashort laser pulse are described in details in [14]. The model consists of two parts: the atomic motion is described within the MD approach, whereas a diffusion differential equation for electrons is accounting for the effect of free carriers via their temperature dynamics (see Eqs. 3 and 4 in [14]). Thus, the combined model provides a detailed atomic-level description of the kinetics of fast non-equilibrium processes, of laser melting with atomic resolution and, at the same time, in continuum ensures for the adequate description of the laser light absorption by conduction band electrons, the energy transfer to the lattice due to electron-phonon coupling, and the fast electron heat conduction (see Fig. 2 in [14]). Based on the atomistic-continuum approach, the MD-TTM model was further developed for the simulation of nanostructuring processes of metal targets in a multiprocessing algorithm (running in parallel) to be applied in large scale modeling tasks. For the case of modeling nanostructuring of thick metal targets with an UV laser pulse on the experimental scale the model is in details in [19, 20], whereas its successful applications for the case of thin targets can be found elsewhere [31, 32].

The schematic view of the total computational cell utilized in the simulation of the nanostructuring process can be seen in Fig. 3. The idea of modeling on the experimental scale is based on symmetry utilization of the experiment described above, see Figs. 1 and 2. The laser beam, sinusoidally shaped in one dimension (Y), leaves a uniform laser intensity distribution in X direction. Dealing with the size of the laser spot of about 40 μm in diameter (flat top in X and sinusoidal shaped in Y), its central part will therefore not be affected by a diffusion process in X direction up to the order of 1 μs of the experimental time. This estimated time is confirmed in preliminary calculations using the TTM approach [15]. Considering this time as long enough for the formation and solidification of the nanostructures [32], this provides a possibility for Periodic Boundary (PB) conditions in X during our simulation time (up to several ns). As a result, since we do not expect any temperature and pressure gradients in the lateral direction parallel to the grating lines (X), for a simulation allowing for identical comparison with experiment, we constructed the computational cell as schematically shown in Fig. 5. The Au MD supercell consisting of 160,000,000 atoms was taken with dimensions of (40×270×200) nm in X, Y, and Z directions respectively. In order to avoid unnecessary and expensive MD integrations in deep bulk of the material, at a certain depth from the surface (in Z direction) we impose Non Reflective Boundary (NRB) conditions. Demanded by the investigated physics, the MD-TTM model was applied only above that limit. The NRB conditions were described in [33] and modified in [32] for the case of a nanostructuring experiment. The NRB conditions, however, are transparent for the heat flux, and the ordinary TTM model was solved beyond the...
NRB taking the electron and phonon temperature dynamics into account on a scale up to 50 µm away from the irradiated surface. Thus, with the described computational cell we simulate a thick Au target with atomic resolution within 200 nm from the surface in the assumption that no phase transition would take place beyond the NRB. This assumption was verified in all our simulations.

The characteristic size of liquid nuclei forming during the laser-induced phase transition process and their initial evolution in the development of the liquid phase has been determined in our previous works [18, 31, 32] to be of the order of 10 nm. For the applied fluence regime therefore, we consider the width of 40 nm in X for the slice of modeled material big enough for a safe application of the PB conditions in that direction. A Free Boundary (FB) is imposed at the top, where the material is irradiated with a laser pulse. Finally, we impose PB in Y direction with the size of MD-TTM domain equal to the period of the sinusoidal pulse shape. In essence, it does not matter where the pulse is centered with respect to the MD domain in Y direction. For convenience, we adjust the maximum peak of the laser intensity to the central point of the sample in Y direction. The subsequent multiple translation of this distribution in X and Y direction will reproduce the intensity profile as it is given by the experimental setup, see Fig. 1.

Fig. 3. Schematic representation of the total computational cell divided to the number of processors \( N_{\text{proc}} \) (256 in this modeling in Y direction) with utilization of the Message Passing Interface (MPI) library (a). In each processor geometry, a combined atomistic-continuum model MD-TTM was realized in 3D-space (based on cut-off distance of the interatomic potential \( r_{\text{cut-off}} \)) subject to find the temperatures of electrons \( T_e \) and phonons \( T_{\text{ph}} \), pressure \( P \), and density \( \rho \) for a given laser source \( S(r,t) \), where \( r \) is the position in space and \( t \) is time. The energy exchange between electrons and phonons \( \Delta E_{\text{e-ph}} \) is accounted for in the model each MD time step. (b). SEM picture (Fig. 2, a) of periodic nanostructures on Au (c) and schematic representation of the computational cell used in the super large scale modeling of this process on the experimental scale (d). PBC are the periodic boundaries. The non-reflective boundaries are transparent for the heat fluxes.

The MD-TTM domain was divided into a 3D mesh with nm³ unit cell and the thermo-physical properties of the atomic subsystem (such as temperature and pressure) were averaged over the closest 26 neighboring cells (that totally accounts for about 2,500 atoms) to suppress their value fluctuations due to MD noise [14, 32]. The choice of the mesh size is justified, on one hand, by its minimization for the efficient neighbor list search and calculations of forces in the MD part. On the other hand, the cell must be big enough to accommodate a group of atoms large enough to introduce their statistical description (under assumption of quasi-equilibrium) for a reliable determination of their macroscopic quantities.

In order to have a possibility for the direct comparison of simulation with experiment, the MD-TTM model has implemented a realistic interatomic potential. For this purpose, we choose the Embedded Atom Method (EAM) implementing the interatomic potential by Zhakhovskii et al. [34], optimized for Au. For an
equilibrium crystal at \( P = 0 \) GPa and \( T = 0 \) K the potential gives \(-367.609 \) kJ/mol for the cohesive energy, 179.4 GPa for the bulk modulus, and 0.4065 nm for the lattice constant. Furthermore, this potential represents the experimental thermophysical properties of the modeled material (such as equilibrium melting temperature, heat capacity, volume of melting, and linear thermal expansion coefficient) with an accuracy of more than 99.5\% [35]. A more detailed description of the model used to describe periodic nanostructuring of metals with UV laser pulse can be found elsewhere [19, 20].

Finally, the computational supercell setup, schematically shown in Fig. 3, was run across 256 processors for 30 days resulting in a simulated experimental time of up to 1000 ps of the nanostructuring process for three different applied incident fluences of 125, 160, and 250 mJ/cm\(^2\). In order to perform such a super large-scale simulation, the MPI (Message Passing Interface library) version of the atomistic-continuum model MD-TTM was adopted, optimized, and used as described in [32]. The processors space geometry 1×256×1 along the \( X, Y, \) and \( Z \) axes was chosen correspondingly for a better load balance during the whole simulation. This choice is governed by our expectation that the laser-assisted material expansion process will proceed upwards and the number of atoms to be processed by each processor therefore will roughly stay the same.

**Results and Discussions**

In general, the mechanism responsible for the nanostructuring process is similar to that described in [31] with a simplified model and later confirmed in a super large scale modeling of nanostructuring of thin [32] and thick metal targets [19, 20]. Namely, the relaxation of the laser-induced compressive stresses generated under conditions of the inertial stress confinement [16, 36] serves as the main driving force responsible for the acceleration of a transiently melted region of the target with subsequent bloating of a nanostructure. Fast electron heat conduction, on the other hand, proceeding by two- or three-dimensional mechanism (depending on the experimental geometry), results in the electron temperature falling significantly below the temperature of atoms in the molten volume. In this case, the electrons, due to electron-phonon energy exchange, provide the conditions for the fast cooling of melted material and rapid solidification of a surface feature generated in the process of hydrodynamic motion of the liquid material.

![Fig. 4. SEM images of the 270 nm periodic structures (a) with the incident fluence \( F_{\text{inc}} \) of 130 mJ/cm\(^2\) (corresponding to the absorbed fluence \( F_{\text{abs}} \) of 87 mJ/cm) are directly compared with the result of an atomistic simulation of the nanostructuring process (b). The atoms are colored with Central Symmetry Parameter (CSP) for distinguishing the atoms with local order (solid – dark blue) from those of disordered structures (grain boundaries – blue, liquid – light blue, surface green, and vapor – red). The magnified view of the final internal microstructure is shown in (c) and (d)](image.png)

The snapshots of atomic configurations, shown in Fig. 4, reveal the formation of dislocation planes during the simulations in \{1,1,1\} directions. Being deviated too much from a perfect Face Center Cubic (FCC) crystalline structure, those atoms, characterized by a higher Central Symmetry Parameter (CSP), appear in light...
blue color, still possess a higher order than that in a liquid, shown by blue color. One can conclude that as a result of thermal expansion of the material in response to the fast laser heating, a number of dislocation planes is induced and propagate in the directions with the lowest energy \{1,1,1\} and eventually form stacking faults. The generated stacking faults have a higher energy barrier for further dislocation planes propagation and stop them as a result. In fact, such a stacking fault organization would reinforce the material, working similarly to that described in short pulse laser peening experiments on steel [37–39] and theoretically studied in semiconductor solids [40].

The resulting snapshots presented in Fig. 5 also illustrate a complex interplay of fast laser melting, rapid cooling, resolidification, and relaxation dynamics of laser-induced stresses in the surface region of a gold target during the nanostructuring process at the incident fluence of 160 mJ/cm². The monocrystalline layer with ultrafine grains (with the size of less than 5 nm) is produced by massive homogeneous nucleation of a solid phase triggered by a deep undercooling of the melted surface region down to below 0.7T_m (below 1000 K) and features random crystallographic orientation of nanograins. The enlarged atomic-scale views of the surface microstructure are shown in Fig. 4 and Fig. 5 reveals a high density of stacking faults and nanograins. Because of the higher fluence as compared to that used in Fig. 4, the internal structure includes a number of solidified features inside of the hollow surface nanobarrier. The formation of similar structural elements has also been observed in frozen nano-spikes generated in the spallation regime, in recent MD simulation by Zhigilei et al [41].

![Fig. 5. Atomistic snapshots taken at the end of modeling at t = 1000 ps and obtained in simulation of the periodic nanostructuring process of thick Au target at the incident fluence of 160 mJ/cm². Similarly to that of Fig. 4, the atoms are colored according to CSP value. A detailed view on the microscopic structure of the internal structures can be seen in magnified boxes.](image)

The computational prediction of the formation of a thin nanocrystalline layer with high density of grains suggests the short pulse laser processing as a viable alternative to the severe plastic deformation [42] or pulsed electrodeposition [43] techniques that are currently employed to yield the maximum structural refinement and high strength of metal samples. Moreover, the localization of the nanocrystalline layer within a thin surface layer (up to about 50 nm deep) of the target and the prominent presence of grain boundaries are two factors that may also increase the ductility of the laser-modified layer [44–46], a property that is generally incompatible with high strength.

As already mentioned above, the fast rate of energy deposition by short laser pulses may lead to the generation of compressive stresses. The laser-generated stresses are particularly high in the regime of stress confinement [16, 36], when the time of the laser heating (defined by the laser pulse duration, \(\tau_p\), or the time of the electron-phonon equilibration, \(\tau_{e-ph}\), whichever is longer) is shorter than the time required for the mechanical relaxation (expansion) of the heated volume, i.e., \(\max(\tau_p, \tau_{e-ph}) \leq L_p/C\), where \(C\) is the speed of sound in the target material and \(L_p\) is the effective depth of the laser energy deposition (optical penetration depth or, in the case of metals, the depth of diffusive/ballistic energy transport during the time of the electron-phonon equilibration). The latter can be a function of electron heat conductivity and for the same pulse duration can significantly alter the value of \(L_p\) depending on the applied fluence [47].

The case of onset of the spallation regime during the nanostructuring process is depicted in Fig. 6 as a sequence of snapshots taken from the simulation with the applied fluence of 250 mJ/cm². The relaxation of the laser-induced stresses formed in the material under conditions of inertial stress confinement [16, 36] induces strong tensile stresses (about ~4GPa) that in the liquid part of the material exceed the material strength (the atmospheric pressure, on the order of 0.1MPa is a negligible contribution here) and promotes the formation of
voids, the precursor of a spallation process [16, 17, 48, 49]. The mechanism of voids formation was investigated in details for metallic and molecular solids in [16]. It was found that if the conditions for inertial stress confinement are fulfilled under the applied irradiation parameters [16, 17, 36], the critical size of voids and the rate of their nucleation are governed by the combination of both quantities of the induced tensile forces (negative pressure) and the atomic temperature. The range at which the characteristic temperature and pressure gradients are established by the time of the electron-phonon equilibrium is related to the effective laser energy deposition depth. It was shown that for short laser pulses (shorter than the characteristic electron-phonon equilibrium time) this depth can be approximated as a function of the material properties at the applied fluence [23, 36, 47, 50, 51]. The growth and percolation of the voids may lead to the separation and ejection of a top layer from the target. By analogy with the term “spallation,” commonly used to describe the dynamic fracture that results from the reflection of a shock wave from the back surface of a sample, the material ejection driven by the relaxation of the laser-induced stresses is commonly referred to photomechanical spallation. In Fig. 6, one can see that at the highest fluence used in this work, the ejecting material is represented by the pre-surface layer opening to the external ambient and forming the features that were observed in the experiment (Fig. 2, c, d).

![Fig. 6. The sequence of atomic snapshots taken at times 50, 200, 500, and 1000 ps, revealing the dynamics of the nanostructure formation at the fluence 250 mJ/cm², that is well above the ablation threshold for Au. The atoms are colored by Central Symmetry Parameter (CSP) parameter](image_url)

Moreover, due to the spatial variation of laser intensity within laser beams that often have Gaussian profiles, both mechanisms of the material ejection photomechanical (large number of clusters and chunks of the material) and photothermal (large number of vapor particles) may coexist and jointly contribute to the material modification induced by the same laser pulse. This is illustrated in Fig. 6, where a number of free volatile atoms contribute to the total material ejected mass. Thus, both photomechanical and photothermal spallation mechanisms are providing channels for effective transformation of the energy deposited by the laser irradiation into material disintegration and vaporization, thus creating conditions for rapid cooling and freezing of the transient liquid structures generated in laser spallation or ablation. Indeed, the temperature of the liquid-crystal interface is already below the melting temperature of Au (measured as \( T = 1000 \) K and less than \( 0.7 T_m \)) by the time of about 600 ps, and the interface is moving up, towards the tip of the nanofeature. As a result, the general picture of the nanostructuring experiment at the fluence above the ablation threshold is like to involve the peel off of a continuous liquid layer with its thickness decreasing towards the center of the laser spot, periodically modulated by energy deposition realized in two-beam interference [19, 20]. Such behavior was also predicted in a mosaic-like reconstruction of the ablation process due to a focused laser spot on silver target presented in [52].

As a result of the presented research, the formation of the final periodic nanostructures on thick Au targets has been investigated both theoretically and experimentally. Fig. 7. In order to simulate the nanostructuring process a hybrid atomistic-continuum (MD-based) super large-scale modeling was applied, which allowed a one-to-one comparison with experiment. The corresponding experimental results were obtained producing an intensity grating on a gold surface with a sinusoidal shape, using single UV-femtosecond laser pulses [19].

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\( t = 50 \) ps

\( X = 40 \) nm

\( Y = 270 \) nm

\( Z = 200 \) nm

\( X \)

\( Y \)

\( Z \)

\( t = 1000 \) ps

\( Z \)

\( Z \)

\( t = 500 \) ps

\( 50 \) nm

\( 400 \) nm

\( 600 \) nm
final surface structures were investigated by SEM while the preparation of a TEM cross section allowed a
detailed look beneath the surface. The high resolution of the latter technique allowed a direct comparison of the
experiment with the results of MD simulation on an atomic scale showing a very good agreement. In Fig. 7, a,
one can see that at the applied fluence of 130 mJ/cm² the resulting snapshot from the atomic simulation reveals
an empty space beneath the surface, that corresponds to the lighten area (platinum) obtained by FIB technique
upon the post experimental treatment of the gold sample. Fig. 7, b, predicts the formation of large number of
monocrystalline structures inside the pre-surface void at the applied fluence of 160 mJ/cm². And, finally, the
highest applied fluence of 250 mJ/cm² (Fig. 7, c), leads to a separation of a thin layer, which opens up to the
surface and, simultaneously with the solidification process, forms the closing nanowalls on the sample surface.

![Figure 7](image-url)

**Fig. 7.** The obtained in simulations final structures are directly compared [19] with the corresponding experimental
SEM images of 270 nm periodic nanostructuring of gold targets with three different average incident fluences
shown in columns for 130 mJ/cm² (a), 160 mJ/cm² (b), and 250 mJ/cm² (c). For the case (a), the obtained
structure was treated by Focused Ion Beam (FIB) method to prepare a cross section, shown in Transmission
Electron Microscopy (TEM) image for visualization of the internal structures in the center of (a). For the case (b),
the areas of the atomic snapshot in black rectangles are zoomed for better visualization of the obtained internal
depositions. The red rectangle on the SEM image in (c) shows the relative position of the
computational cell when simulating the nanostructuring process. In all atomic snapshots, taken at the final time of
1000 ps, the atoms are colored according the CSP indicating gaseous atoms (red), surface (green), liquid
ambient (light blue), dislocation planes and defects (blue), and the atoms with crystalline surrounding (dark blue)

**Conclusion**

Rapid expansion of the domain of practical applications of ultrashort pulse lasers into new areas where
precise nanoscale control of the material modification is required can only be sustained with an improved
fundamental understanding of laser-materials interactions. The ability to achieve high spatial resolution with UV
laser pulses in material modification and microstructure engineering usually relies on fast and highly localized
energy deposition that, unavoidably, creates the conditions of strong electronic, thermodynamic, and mechanical
nonequilibrium.

Theoretical and computational description of the materials behavior far from the equilibrium is challenging and requires the revision of existing (or development of new) theoretical models and computational
approaches. In particular, the intricate connections between the ultrafast material response to the laser excitation,
generation of crystal defects, phase transformations, and mass transfer occurring under highly nonequilibrium
conditions can only be accounted for by combining the research expertise and theoretical approaches. Close
interdisciplinary collaborations and innovative design of new multiscale approaches (such as MD-TTM model)
incorporating dissimilar physical concepts and operating at different time- and length-scales are needed for
making a significant progress in the theoretical and computational description of laser-material interactions.
The presented work revealed one of the concepts of the internal structures formation on the example of nanocrystalline silver. The structures obtained in the periodic nanostructuring experiments with short UV laser pulses. The applied MD-based model allowed for the direct comparison of the simulated and experimental results and revealed their close match. The demonstrated approach, therefore, opens the way to predict properties of surface modifications for industrial applications. This fact emphasizes that the understanding of these processes on an atomic scale and being able to precisely simulate them provides a powerful tool for designing new nano-technology applications.

**References**


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Dmitry S. Ivanov graduated from Physico-Technical Department of Peter the Great St. Petersburg Polytechnic University (Politech) in 1997 with specialization of “Space Research”. In 2004 he obtained his PhD degree in the University of Virginia (USA). At present he works as a Senior Researcher at Technical University of Kaiserslautern and University of Kassel (Germany). Also, since 2015 he has been a co-chief of the International Laboratory of “Lasers Micro- and Nanotechnologies” at Saint Petersburg State University of Information Technologies, Mechanics and Optics (Russia). His scientific research interests cover modeling of short laser pulse melting, spallation, ablation processes as well as laser-induced nanostructuring of metals, semiconductors, and dielectric surfaces.

Андреас Блюменштайн получил степень магистра в Университете Гёттингена в 2013 г. Работал над проблемами откатализ и самоорганизующихся наноструктур. В настоящее время – аспирант Лазерной лаборатории Гёттингена и Университета города Кассель (Германия). Область научных интересов включает в себя лазерное структурирование поверхностей материалов и нелинейную оптику.

Andreas Blumenstein received a Master of Science degree from the University of Göttingen in 2013 working on photo-catalysis and self-assembled nano-structuring. He is currently doing his PhD at the Laser-Laboratorium Göttingen e.V. and the University of Kassel. Scientific interests are: laser surface structuring, nonlinear optics, solid state physics.
Barbel Rethfeld is a Heisenberg Professor of Applied Theoretical Physics at the University of Kaiserslautern since 2013. She received her PhD degree from the University of Braunschweig in 1999 and has worked since then at the University of Essen, Germany, L.D. Landau Institute for Theoretical Physics in Moscow, Russia, and GSI Helmholtz Institute in Darmstadt, Germany. In 2007 she established her own research group at the University of Kaiserslautern, after she was awarded an Emmy Noether grant from the German Research Foundation (DFG). Her research interest is in the theoretical description of ultrafast dynamics of laser-excited solids with an emphasis on nonequilibrium processes.

Vadim P. Veiko – D.Sc., Professor, Head of Laser Technologies and Systems Department of Saint Petersburg State University of Information Technologies, Mechanics, and Optics (ITMO University). He is an Honored scientist of the Russian Federation, USSR State Prize winner in the field of science (1986), RF Government Prize winner in the field of education (2010). He is a member of Program Committee for a number of international conferences «Fundamentals of Laser Assisted Manufacturing and Nanotechnologies» (FLAMN). His scientific research interests cover such fields like Laser-Matter Interaction, Fundamentals of Laser Technologies, etc. He is the author and coauthor of more than 300 scientific publications, among them 12 monographs in Russian, English and Chinese.

Evgeniy B. Yakovlev – D.Sc., Professor of Laser Technologies and Systems Department of Saint Petersburg State University of Information Technologies, Mechanics and Optics (ITMO University). He has graduated from Leningrad Institute of Fine Mechanics and Optics. In 1985 he obtained his PhD degree with specialization of "Physical Electronics including Quantum One" and in 1999 he presented his habilitational thesis with specialization of "Optical and Optoelectronic Devices". He was awarded with the Prizes of the Presidium of the Academy of Sciences of the USSR for the best scientific research in the direction of "Fundamental Problems of Microelectronics" in 1976, the USSR Ministry of Higher Education Institutions for the best scientific research in 1983, the Government of the Russian Federation in the field of education "Creation of innovative scientific educational system for key personnel training in the area of material processing laser technology" in 2010. He is a member of Program Committee for a number of international conferences «Fundamentals of Laser Assisted Micro- and Nanotechnologies» (FLAMN). His scientific interests are: Laser-Matter Interaction, Laser Technologies. He is the author of more than 160 scientific and methodology works, patents. He is an Editor-in-Chief of the magazine "News of Higher Educational Institutions. Instrumentation", RAS Expert.
Мартин Е. Гарсия — профессор Университета г. Кассель, отделение математики и естественных наук. Окончил институт Балсеиро (Барилоджи, Аргентина) в 1984 г. по специальности «Физика» и в 1992 г. защитил кандидатскую диссертацию (PhD) в Берлинском Университете. После работы младшим научным сотрудником (PostDoc) в Соединенных Штатах Америки и Испании вернулся в Берлинский Университет, где до 1999 г. проходил докторантуру на кафедре теоретической физики. Впоследствии занимал временные позиции профессора в Университете г. Грейсвальд (Германия) и пригласенного профессора в Испании и Мексике. В 2004 г. получил постоянную позицию профессора на отделении теоретической физики Университета г. Кассель (Германия). Там же с 2006 по 2008 гг. профессор Гарсия занимал должность заместителя декана на отделении естественных наук. Область его научных интересов обширна и включает в себя изучение электронных, оптических и диффузионных свойств твердых тел и наноструктур, процессов фазовых переходов, индукционных лазерных излучений, биофизических процессов свертывания белков и межклеточной коммуникации.

Martin E. Garcia has studied Physics at the Instituto Balseirlo, Bariloche, Argentina. He obtained his PhD in Physics at the Free University of Berlin in 1992. After PostDoc position in the USA and Spain he returned to the Free University, where he obtained his Habilitation in Theoretical Physics in 1999. A temporal appointment as a Professor at the University of Greifswald, Germany, followed. After positions as invited Professor in Spain and Mexico he obtained a permanent appointment as a Professor for Theoretical Physics at the University of Kassel, Germany in 2004. He was a Vice Dean of the Department of Natural Sciences between 2006 and 2008. His research interests are quite broad, ranging from electronic, optical and transport properties of solids and nanostructures, interaction of lasers with materials, laser induced phase transitions, up to Biophysics (protein folding and intercellular communication).

Петер Симон — руководитель отдела “Коротких импульсов и наноструктур” Лазерной Лаборатории Геттингена, Германия. В целом, его исследования направлены на применение сверхкоротких лазерных импульсов в создании экстремальных концентраций энергии в пространстве и времени. Подобные источники энергии позволяют генерировать крайне высокие интенсивности при достаточно умеренной доставляемой энергии и с высокой степенью контроля. Благодаря этому, сверхкороткие лазерные импульсы образуют основной подход для применения в технических и технологических пакетах. Предмет его текущих исследований включает в себя генерацию и концентрирование лазерных импульсов с высокой энергией и интенсивностью, а также субмикронная структурирование поверхностей технологических материалов.

Peter Simon is a head of the Department ‘Short Pulses / Nanostructures’ at the Laser Laboratory Goettingen, Germany. His research in general is devoted to the usage of ultrashort lasers pulses and their ability for extreme concentration of energy in time and space. Such sources allow the generation of very high intensities while delivering only moderate energies in a well-controlled manner. Therefore the ultrashort-pulse laser technology holds great potential for applications in a generation and compression of energetic few-cycle pulses, and the submicron-scale surface texturing of technical materials.

Юрген Илеман — старший научный сотрудник Лазерной Лаборатории Геттингена, Германия. Защитил диплом по специальности физика в 1984г. и в 1987г. получил ученую степень PhD по физической химии в университете Геттингена. С 1984 по 1988 года работал в Институте биофизической химии Макс-Планка в Йеттингене, где занимался приготовлением короткоколлампальным лазерным излучением, что остается основным направлением исследований Юргена Илемана по настоящее время. Исследование таких структур на поверхности металлов, полупроводников, диэлектриков или полимеров может приводить к получению материалов с новыми уникальными свойствами или с особыми электрическими или механическими характеристиками. В зависимости от свойств материала и морфологии получаемых структур, могут быть реализованы новые устройства такие как: микро-лазеры, оптические нано-переключатели, оптические устройства хранения информации, био-сенсоры или генераторы защиты прав производителя. Более того, получаемые наноструктурированные поверхности могут быть использованы для уменьшения потерь вызываемых отражением, для изменения смачиваемости, в биотехнологиях или в ювелирных работах.

Juergen Ihlemann is a senior research scientist at the Laser Laboratory Goettingen, Germany. He received his diploma in physics (1984) and Ph.D. in physical chemistry (1987) from the University of Göttingen. From 1984 to 1988 he was with the Max-Planck-Institute for biophysical chemistry, Göttingen, where he was working on picosecond laser spectroscopy. In 1989 he joined Laser-Laboratorium Göttingen, where the study of ultrashort laser pulse generated periodic nanostructures became one of the main topics of his present research. Such structures on the surface of metals, semiconductors, dielectrics or polymers can generate new unique material properties with very special electrical or mechanical characteristics. Depending on the specific material parameters and the morphology of the structures, new devices like micro-lasers, optical nano-switches, optical storage devices, bio-sensors or anti-fraud features can be realized. Furthermore, surface textures can be used to improve the tribological properties of special tools, for the reduction of reflection losses, to modify the wettability or the cell growth properties or as decoration elements for the refinement of precious goods.